

The Crystal Structure of Tugtupite - a New Mineral, $\text{Na}_8\text{Al}_2\text{Be}_2\text{Si}_8\text{O}_{24}(\text{Cl},\text{S})_2$

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The structure of the new mineral tugtupite $\text{Na}_8\text{Al}_2\text{Be}_2\text{Si}_8\text{O}_{24}(\text{Cl},\text{S})_2$ has been determined by single-crystal X-ray methods. The crystals are tetragonal with space group $I\bar{4}$. They are strongly piezoelectric but show no pyroelectric effect. Optically the mineral is orthorhombic with values of $2V_\gamma$ up to 10° and $n_\alpha \approx n_\beta = 1.496 \pm 0.001$, $n_\gamma = 1.502 \pm 0.002$. The specific gravity is 2.30 ± 0.02 . The unit-cell constants are $a = 8.583 \pm 0.004$, $c = 8.817 \pm 0.004$ Å. Atomic and thermal vibrational coordinates were refined to $R = 8.9\%$ by full-matrix least-squares methods and with three-dimensional data. The following bond lengths were found:

$$\begin{aligned}\text{Be-O} &= 1.608 \pm 0.012 \text{ \AA} \\ \text{Al-O} &= 1.762 \pm 0.014 \\ \text{Si-O} &= 1.601 \pm 0.008\end{aligned}$$

The last value is an average of four independent determinations.

Introduction

The mineral was described by Sørensen (1960, 1962) under the name of beryllium sodalite, and by Semenov & Bykov (1960), who used the name beryllio-sodalite. Its chemical and physical properties turned out to be so different from those of sodalite that a distinct name was justified. Tugtupite was suggested after the place where the mineral was originally found (Tugtup agtakõrfia in the Ilimaussaq intrusion, South Greenland), and this name was approved in 1965 by the I.M.A. Commission on New Minerals and Mineral Names.

The crystals are white when kept in darkness, but change to pink when exposed to daylight. Optically the mineral is orthorhombic with values of $2V_\gamma$ up to 10° .

$$\begin{aligned}n_\alpha \approx n_\beta &= 1.496 \pm 0.001 \\ n_\gamma &= 1.502 \pm 0.002; c \neq \gamma.\end{aligned}$$

The cleavage is good parallel to $\{110\}$ and $\{101\}$. (A more extensive crystallographic description will be given in *Medd. om Grønland*, vol. 181, No. 4).

Experimental

Chemical and physical properties

The result of a chemical analysis is given in Table 1. The formula of the mineral was calculated to be $\text{Na}_8\text{Al}_2\text{Be}_2\text{Si}_8\text{O}_{24}(\text{Cl},\text{S})_2$ with traces of K and Mg substituted for Na, the S content being about 5% of the Cl content.

Table 1. Chemical analysis of tugtupite

| | Weight % | Metal atoms in 100 g | Metal atoms in $100 \cdot \frac{12.00}{1.295}$ g† |
|-------------------------|----------|-------------------------|---|
| SiO_2 | 51.58 | 0.861 | 7.98 |
| Al_2O_3 | 11.15 | 0.219 | 2.03 |
| Na_2O | 25.52 | 0.822 | 7.62 |
| BeO | 5.40 | 0.216 | 2.00 |
| Cl | 7.28 | 0.205 | 1.90 |
| S* | 0.33 | 0.010 | 0.09 |
| K_2O | 0.12 | 0.002 | 0.02 |
| MgO | 0.20 | 0.005 | 0.05 |

* Present as S^{2-} .

† The sum of Be, Al and Si was 1.295 in 100 g; in sodalite the sum of Al and Si is 12. The figures in the third column were therefore multiplied by the ratio $\frac{12.00}{1.295}$ to give the last column.

Table 2. Observed and calculated values of $\sin^2 \theta$ up to 0.2500

| $\sin^2 \theta_{\text{obs}}$ | Intensity | Indices | $\sin^2 \theta_{\text{calc}}$ |
|------------------------------|-----------|---------|-------------------------------|
| 0.0158 | s | 101 | 0.0157 |
| 0.0162 | vw | 110 | 0.0161 |
| 0.0306 | w | 002 | 0.0306 |
| 0.0323 | vvw | 200 | 0.0323 |
| 0.0467 | m | 112 | 0.0467 |
| 0.0480 | vs | 211 | 0.0480 |
| 0.0630 | vvw? | 202 | 0.0628 |
| 0.0647 | vvw | 220 | 0.0645 |
| 0.0770 | vw | 103 | 0.0769 |
| 0.0803 | vw | 301 | 0.0802 |
| 0.0808 | vvw | 310 | 0.0807 |
| 0.0952 | m | 222 | 0.0951 |
| 0.1093 | w | 213 | 0.1091 |
| 0.1113 | w | 312 | 0.1112 |
| 0.1126 | w | 321 | 0.1125 |
| 0.1384 | w | 114 | 0.1384 |
| 0.1413 | w | 303 | 0.1414 |
| 0.1451 | m | { 330 | 0.1452 |
| | | { 411 | 0.1448 |
| 0.1595 | vw | 402 | 0.1596 |
| 0.1612 | vvw | 420 | 0.1613 |
| 0.1738 | vw | 323 | 0.1737 |
| 0.1868 | vvw | 224 | 0.1869 |
| 0.1918 | vw | 422 | 0.1919 |
| 0.2026 | vvw? | 314 | 0.2030 |
| 0.2062 | vw | 413 | 0.2059 |
| 0.2093 | w | 431 | 0.2093 |
| 0.2314 | vvw | 215 | 0.2315 |
| | | { 521 | 0.2416 |
| 0.2410 | vvw? | { 512 | 0.2403 |

The specific gravity, as determined by hydrostatic weighing, is 2.30 ± 0.02 , while the density calculated from the chemical analysis and the unit-cell volume is 2.36 g.cm^{-3} . By means of differential thermal analysis it was ascertained that no reaction takes place below 630°C . X-ray powder diagrams showed no change in symmetry when the mineral was heated to the melting point ($\sim 950^\circ\text{C}$).

The crystals are strongly piezoelectric, but no pyroelectric effect could be detected.

X-ray data

The axes of the unit cell were determined at room temperature by the Guinier technique with Cu $K\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$):

$$a = b = 8.583 \pm 0.004 \text{ \AA}$$

$$c = 8.817 \pm 0.004 \text{ \AA}$$

$$V = 658 \text{ \AA}^3$$

$$Z = 1$$

Table 2 gives the observed and calculated values of $\sin^2 \theta$ with their roughly estimated intensities. NaCl has been used as a standard, $a = 5.6400 \text{ \AA}$.

The symmetry of precession and Weissenberg films and the strong piezoelectric effect reduced the possible space groups to $I4$ and $I\bar{4}$. The only way of deciding between these two with certainty would be detection of a positive pyroelectric effect, which is allowed only for $I4$. The close relation with the sodalite structure (space group $P43m$ or $P43n$, see later) and the types of equivalent positions in the two space groups strongly favour $I\bar{4}$, and consequently the structure determination was started on the assumption of this space group. The final agreement between calculated and observed structure factors showed that the choice was correct.

Intensity data

An approximately equidimensional crystal ($\sim 0.3 \text{ mm}$) was rotated about the c axis, and the reciprocal layers $hk0-hk6$ were recorded with unfiltered Cu radiation and the Weissenberg technique.

Another crystal ($\sim 0.4 \times 0.1 \text{ mm}$), ground to ellipsoidal shape, was rotated along the long axis of the ellipsoid, which was nearly parallel to the b axis of the crystal, and the $h0l-h6l$ reflexions were photographed. As the latter crystal turned out to be not quite perfect, only the equatorial reflexions were measured. The intensities were estimated visually by means of a calibrated strip. The multiple film technique was used and the conversion factors between the layers were determined experimentally. The exposure times for the packets were chosen so that one film from one packet should be more or less identical with a film in another packet.

As $I_{hkl} = I_{\bar{h}\bar{k}l}$ the same reflexions were measured on several films and on different positions within the film; in many cases β reflexions also gave valuable information. Due regard was paid to all measurements.

In case of $\alpha_1 - \alpha_2$ splitting a weighted mean was calculated. The fourfold symmetry was checked and found to be fulfilled within the experimental error. (Optically, however, the symmetry is lower; see above.) Mean values of I_{hkl} and $I_{\bar{h}\bar{k}l}$ were used when different. The intensities were corrected for Lorentz and polarization factors, but not for absorption; see later*.

Structure determination and refinement

The two Patterson projections $P(uv)$ and $P(uw)$ were in good agreement with what could be expected from knowledge of the sodalite structure: Cl in 000, Al and Be in $0\frac{1}{2}\frac{1}{4}$ and $0\frac{1}{2}\frac{3}{4}$ (only the refinement could decide the distribution between the two positions; see later), and Si in a general position with $xyz \approx 0\frac{1}{4}\frac{1}{2}$, the 12 metal atoms Be, Al, Si thus occupying the positions of Al and Si in sodalite. As Cl is in 000 the Patterson projections bear great resemblance to electron-density projections, and some information could be obtained about the other atoms. Two sets of O atoms stood out, but the last O was overlapped by Na and again use must be made of the sodalite structure, whose 24 O atoms can be divided into three groups corresponding to the three eightfold general positions in $I\bar{4}$. As the cubic space groups have cyclic rotation of the coordinates, a well resolved O atom in a projection will give information about the third coordinate of another O atom.

As the electron-density projection along the c axis is centrosymmetric the corresponding difference maps were calculated until R was 23%.

A minimum residual program (Bhuiya & Stanley, 1963) written by J. Danielsen refined the coordinates to $R = 14\%$ and $R = 9\%$ for the $hk0$ and $h0l$ reflexions respectively. A common isotropic B value of 1.0 was used, and no attempt was made to vary the temperature factor at this stage. In order to calculate F_{h0l} Al and Be had to be placed in the twofold positions. Both possibilities were tried; the one with Be in $0\frac{1}{2}\frac{1}{4}$ and Al in $0\frac{1}{2}\frac{3}{4}$ converged more rapidly and was therefore used for the three-dimensional R -minimization later on. (The final least-squares calculations proved this choice to be correct; see below.)

Data from the reciprocal layers $hk0-hk6$ were put on the same scale using the $h0l$ reflexions. For a given l the ratio between common F values from the two zones was not constant but varied with ξ because of absorption in the larger crystal used for recording the $hk0-hk6$ reflexions. The ratios were extrapolated to $\xi = 0$. These preliminary values for the conversion factors between the layers were later refined in the usual way using $F_{\text{obs}}/F_{\text{calc}}$ for each layer when more accurate coordinates were known. There seems to be a possibility of extinction, but no correction for this has been made. After a readjustment of the conversion factors the final

* No allowance was made for spot shape distortion other than averaging the intensities of reflexions from the upper and lower half of the film.

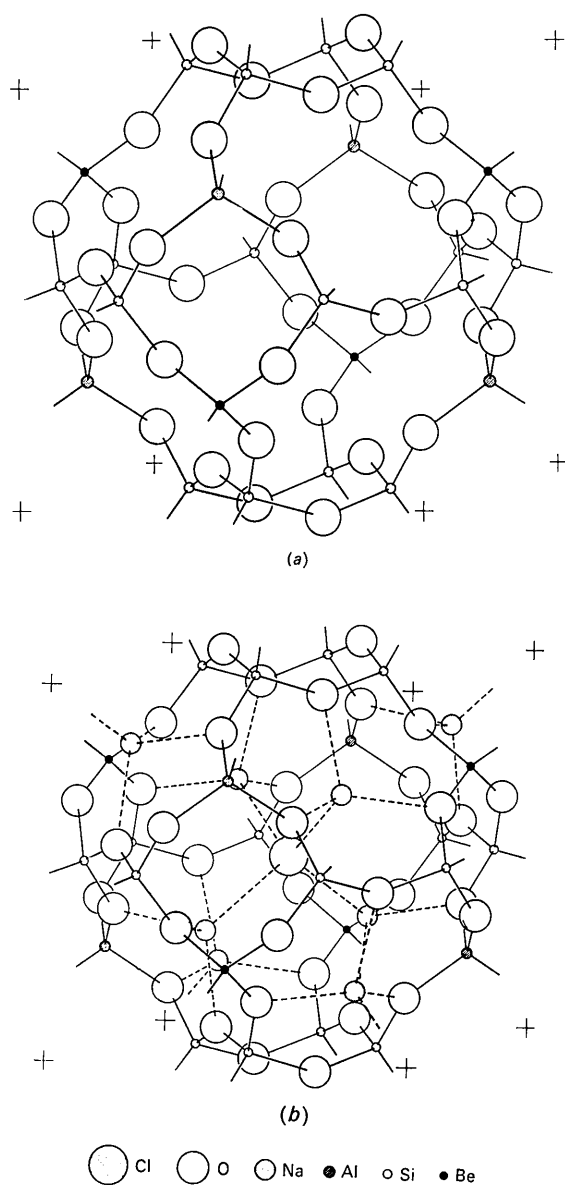


Fig. 1. Modified clinographic projection of tugtupite. $\varphi = 9^\circ 28'$ is the angle of elevation in the a - c plane of the lines of projection after the rotation $\theta = 18^\circ 26'$ about the c axis. In this special kind of projection the b and c axes remain in the plane of the paper and mutually perpendicular. The a axis points towards the reader to the left. The corners of the unit cell are indicated by crosses. In (a) only the Si, Be, Al and O are shown, while the complete structure is given in (b).

refinement of the parameters was done by using a full-matrix least-squares program (Busing, Martin & Levy, 1962). After two cycles new scale factors were calculated and three more cycles were computed until the shifts were less than 1% of the standard deviations. The final R was 8.9%. The weighting function was of the type

$$w = A_1^2 \quad F_o \leq 2.0$$

$$w = A_2^2 \cdot F_o^{2B_2}; \quad F_o > 2.0$$

with

$$A_1 = 5.0$$

$$A_2 = 10; B_2 = -1; F_{\min} \approx 2.0$$

All possible reflexions were observed, except five. These were not included in the least-squares calculations. f values were taken from *International Tables for X-ray Crystallography* (1962, pp. 202-3), and values for neutral atoms were used. Only isotropic temperature factors were introduced and refined with the positional parameters.

Results and conclusions

The observed and calculated structure factors are given in Table 3 as ten times the absolute values. The final atomic positions with their Debye temperature factors are found in Table 4.

As mentioned above, the crystal is not small enough for the absorption to be negligible ($\mu \sim 86 \text{ cm}^{-1}$). The B values determined by the least-squares method are consequently too low. Especially the $hk0$ reflexions seem to be affected.

The structure consists of linked MO_4 tetrahedra with $M = \text{Be, Al}$ and Si as the metallic cations (Fig. 1). These tetrahedra share corners, Cl and Na being placed in holes in the framework. Characteristic are the 8-rings with alternating metal and oxygen atoms. One third of the rings contain only Si, while the rest have Be, Si and Al alternating. Fig. 1(a) shows the arrangement of the 8-rings. Fig. 1(b) gives the complete structure. As Be and Al are placed on a fourfold axis the BeO_4 and AlO_4 tetrahedra are regular. The metal-oxygen distances are $1.608 \pm 0.012 \text{ \AA}$ and $1.762 \pm 0.014 \text{ \AA}$ for Be-O and Al-O respectively. The SiO_4 tetrahedra, however, are distorted. Of the four O atoms in the tetrahedron, two are bound to other Si atoms, one to Be and one to Al. The bond lengths with their standard deviations are given in Table 5. The Si-O distances in the pure rings are 1.646 (A) and 1.611 (B) alternating, while the mixed rings have distances of 1.588 (C) when O is linked to Be and 1.566 (F) when linked to Al.

Table 4. Atomic coordinates and thermal parameters with their standard deviations

| | x/a | $\sigma(x/a)$ | y/b | $\sigma(y/b)$ | z/c | $\sigma(z/c)$ | B | $\sigma(B)$ |
|------|--------|---------------|--------|---------------|--------|---------------|------|-------------|
| O(1) | 0.1471 | 0.0008 | 0.1332 | 0.0008 | 0.4431 | 0.0013 | 0.63 | 0.14 |
| O(2) | 0.3467 | 0.0009 | 0.0362 | 0.0009 | 0.6512 | 0.0015 | 0.86 | 0.13 |
| O(3) | 0.4261 | 0.0008 | 0.1506 | 0.0008 | 0.1347 | 0.0014 | 0.79 | 0.13 |
| Na | 0.1575 | 0.0005 | 0.1970 | 0.0005 | 0.1815 | 0.0008 | 1.12 | 0.10 |
| Si | 0.0134 | 0.0003 | 0.2535 | 0.0003 | 0.4956 | 0.0005 | 0.22 | 0.08 |
| Be | 0.0000 | — | 0.5000 | — | 0.2500 | — | 0.66 | 0.41 |
| Al | 0.0000 | — | 0.5000 | — | 0.7500 | — | 0.26 | 0.10 |
| Cl | 0.0000 | — | 0.0000 | — | 0.0000 | — | 1.50 | 0.11 |

Table 5. Bond lengths with their standard deviations

| Bond | M-O Distance | σ |
|----------|--------------|----------|
| Si-O | A 1.646 Å | 0.014 Å |
| Si-O | B 1.611 | 0.013 |
| Si-O | C 1.580 | 0.018 |
| Si-O | D 1.566 | 0.017 |
| Average: | 1.601 | 0.008* |
| Al-O | E 1.762 | 0.014 |
| Be-O | F 1.608 | 0.012 |

* Taken as the average of σ divided by the square root of the numbers of independent determinations, as suggested by Smith & Bailey (1963).

Smith & Bailey (1963) give as best values for Si-O and Al-O 1.61 Å and 1.75 Å respectively, and our values agree within the experimental error. Only a few Be-O distances have been reported in the literature, but the value found in tugtupite falls within the interval given by *International Tables for X-ray Crystallography* (1962, p.260). The question of the space group of sodalite, $\text{Na}_8\text{Al}_6\text{Si}_6\text{O}_{24}\text{Cl}_2$, is still open. In the paper of Pauling (1930) it was given as $P\bar{4}3n$, while Barth (1932) argued for $P\bar{4}3m$. Later Saalfeld (1961) discussed the matter in relation to other members of the sodalite group and proposed the space group $P\bar{4}3n$. With the technique used in the early structure determinations it was not possible to determine the Si-O and Al-O distances with sufficient accuracy to decide whether the 6 Al and 6 Si are in two different special positions ($P\bar{4}3n$) or statistically distributed in a 12-fold position ($P\bar{4}3m$).

In view of the ordered arrangement of the cations of tugtupite it seems worth while to re-examine the

structure of sodalite. We hope to be able to do so in the near future.

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On the Equivalence of Structure Invariants*

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The triangle in reciprocal space associated with a given structure invariant is defined and the concept of the equivalence of structure invariants is introduced. It is assumed that many of the interatomic triangles (especially the smaller ones) in a crystal structure are congruent to each other and that the triangles in each collection of congruent interatomic triangles are more or less randomly oriented in space. Under these assumptions the values of equivalent structure invariants are approximately equal, especially if the sides of the associated triangle are small. This result facilitates the solution of the problem of direct phase determination for structures satisfying our assumptions.

1. Introduction

The structure invariants are those entities whose values are uniquely determined by a crystal structure. They play a fundamental role in the problem of direct phase

determination. An important set of structure invariants consists of the linear combinations of the phases

$$\varphi_1 + \varphi_2 + \varphi_3, \quad (1.1)$$

in which the abbreviations

$$\varphi_i = \varphi_{h_i}, \quad i = 1, 2, 3, \quad (1.2)$$

have been used and the assumption

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